

Synthesis of Hexagonal Nanophases in the $\text{La}_2\text{O}_3 - \text{MO}_3$ ($\text{M} = \text{Mo}, \text{W}$) Systems

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ABSTRACT

We report a study of nanophases in the $\text{La}_2\text{O}_3 - \text{MO}_3$ ($\text{M} = \text{Mo}, \text{W}$) systems, which are known to contain a variety of good oxygen-ion and proton conductors. Mechanically activated $\text{La}_2\text{O}_3 + \text{MO}_3$ ($\text{M} = \text{Mo}, \text{W}$) mixtures have been characterized by DSC and XRD with Rietveld refinement, the microstructure of the materials has been examined by SEM, and their conductivity in dry and wet air has been determined using impedance spectroscopy. In both systems, the formation of hexagonal $\text{La}_{15}\text{M}_{8.5}\text{O}_{48}$ (phase II, 5H polytype) ($\text{M} = \text{Mo}, \text{W}$) nanophases is observed for composition 1:1, with exothermic peaks in the DSC curve in the range $\sim 480 - 520^\circ\text{C}$ for $\text{La}_{15}\text{Mo}_{8.5}\text{O}_{48}$ and $\sim 685 - 760^\circ\text{C}$ for $\text{La}_{15}\text{W}_{8.5}\text{O}_{48}$. The crystallite size of the nanocrystalline tungstates is ~ 40 nm and that of the nanocrystalline molybdates is ~ 50 nm. At higher temperatures ($\sim 630 - 690$ and $\sim 1000^\circ\text{C}$), we observe irreversible reconstructive phase transitions of hexagonal $\text{La}_{15}\text{Mo}_{8.5}\text{O}_{48}$ to tetragonal $\gamma - \text{La}_2\text{MoO}_6$ and of hexagonal $\text{La}_{15}\text{W}_{8.5}\text{O}_{48}$ to orthorhombic $\beta - \text{La}_2\text{WO}_6$. We compare temperature dependences of conductivity for nanoparticulate and microcrystalline hexagonal phases and high-temperature phases differing in density. Above 600°C , oxygen ion conduction prevails in the coarse-grained $\text{La}_{18}\text{W}_{10}\text{O}_{57}$ (phase I, 6H polytype) ceramic. Low-density $\text{La}_{15}\text{W}_{8.5}\text{O}_{48}$ and $\text{La}_{15}\text{Mo}_{8.5}\text{O}_{48}$ (phase II, 5H polytype) nanoceramics exhibit predominantly electron conduction with an activation energy of 1.36 and 1.35 eV, respectively, in dry air.

Particle size and morphology of starting oxide powders and mechanically activated powders

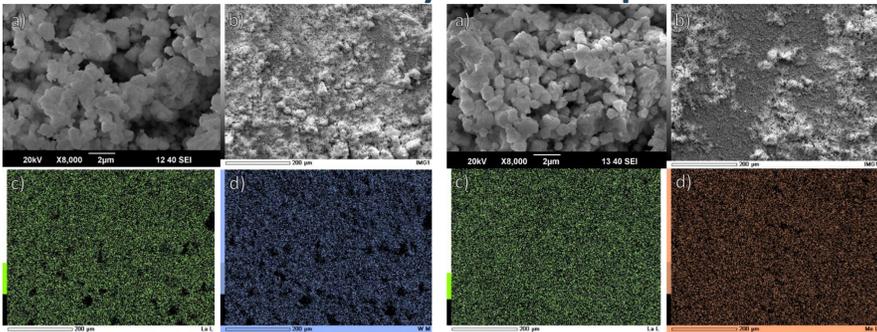


Fig. 1 SEM image (a) (magnification 2 μm) and (b) (magnification 200 μm), (c, d) La and M X-ray maps of the mechanically activated (m/a) $\text{La}_2\text{O}_3 + \text{MO}_3$ oxide mixture. W (left), Mo (right)

Thermal analysis

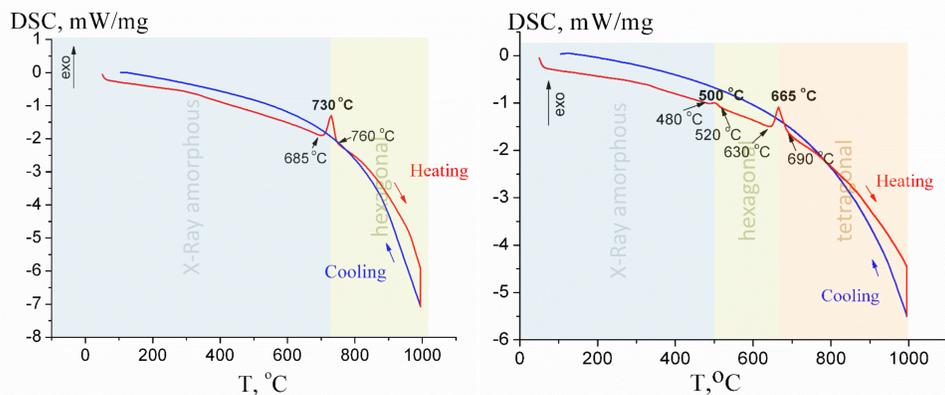


Fig. 2 Data obtained in a DSC cell during heating to 1000°C , illustrating the exothermic process in a mechanically activated $\text{La}_2\text{O}_3 + \text{MO}_3$ mixture. $\text{M} = \text{W}$ (left), Mo (right)

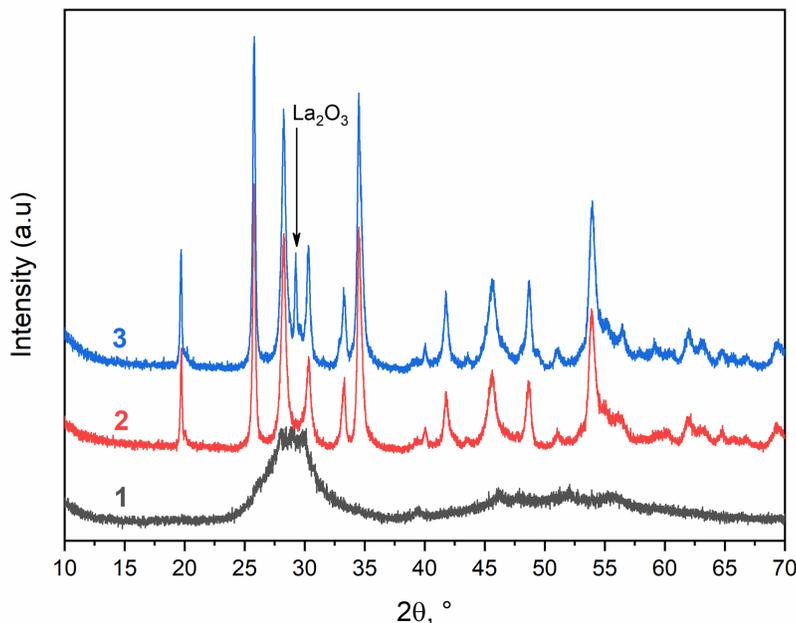


Fig. 3 XRD patterns of mechanically activated (m/a) $\text{La}_2\text{O}_3 + \text{WO}_3$ mixture (1) after heating in DSC cell up to 685°C , (2) after heating in DSC cell up to 760°C , (3) after thermal annealing at 600°C for 96 h. La_2O_3 (ICDD PDF 1523968) - 41.1(6) wt.%.

X-Ray Powder Diffraction

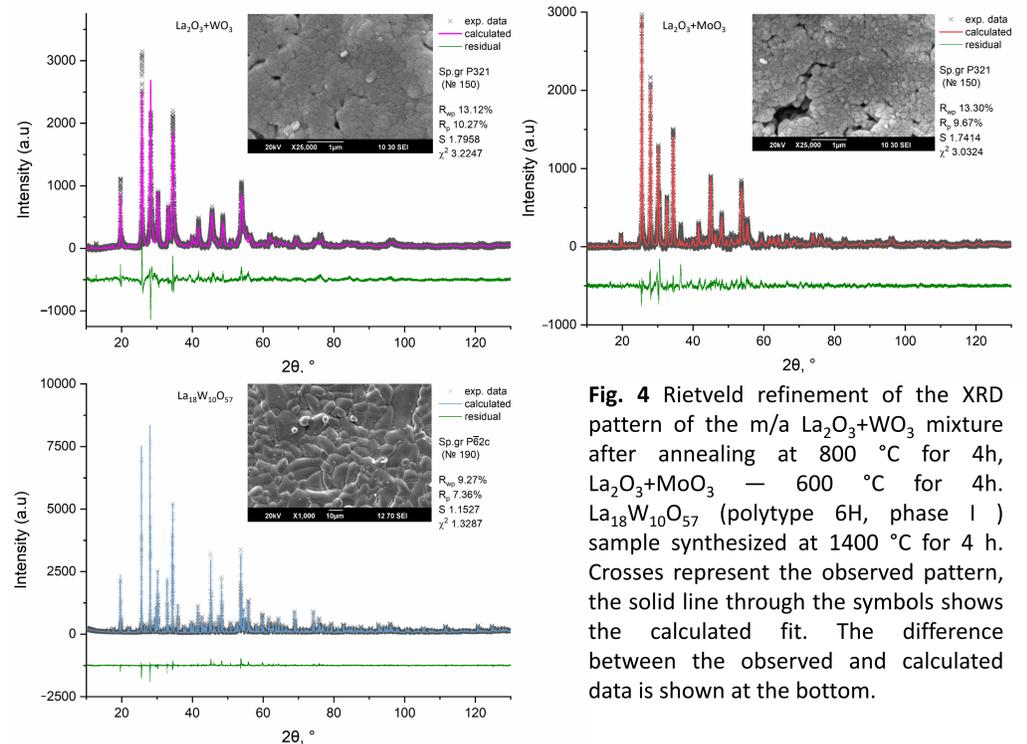


Fig. 4 Rietveld refinement of the XRD pattern of the m/a $\text{La}_2\text{O}_3 + \text{WO}_3$ mixture after annealing at 800°C for 4h, $\text{La}_2\text{O}_3 + \text{MoO}_3 - 600^\circ\text{C}$ for 4h. $\text{La}_{18}\text{W}_{10}\text{O}_{57}$ (polytype 6H, phase I) sample synthesized at 1400°C for 4 h. Crosses represent the observed pattern, the solid line through the symbols shows the calculated fit. The difference between the observed and calculated data is shown at the bottom.

Ionic conductivity

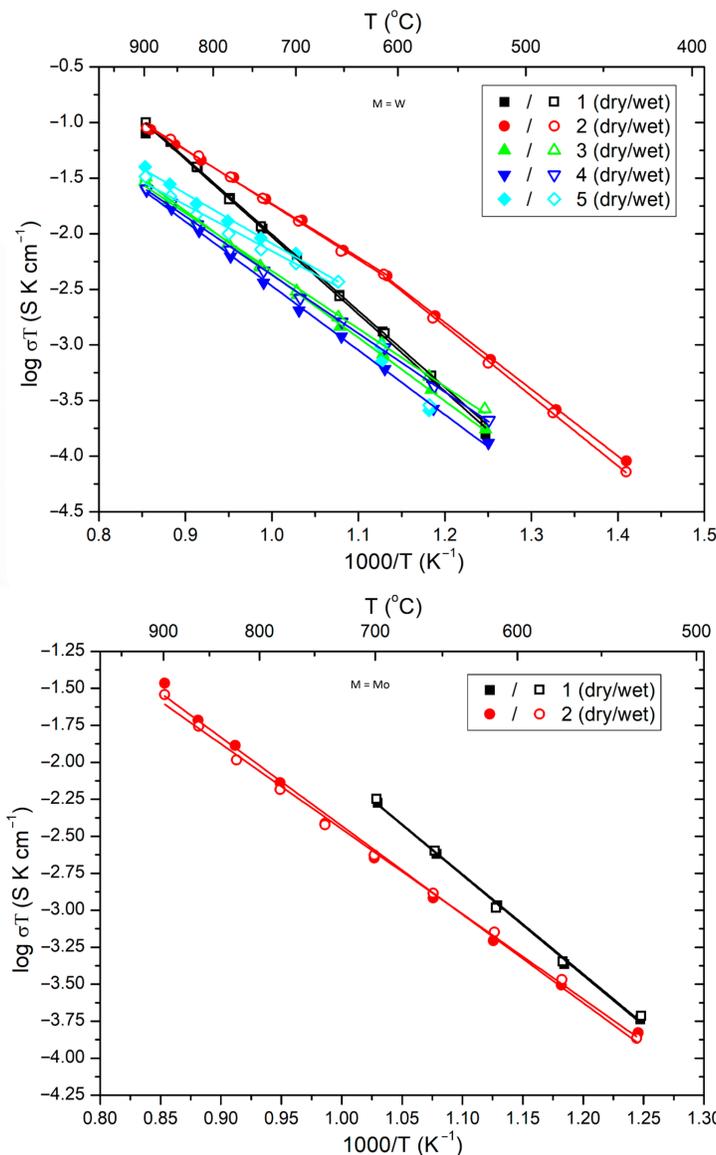


Fig. 5 Arrhenius plots of total conductivity for LaMO materials in dry (filled data points) and wet (open data points) air: **Top** (1) low-density hexagonal $\text{La}_{15}\text{W}_{8.5}\text{O}_{48}$ nanoceramic (relative density of 60.0%), (2) dense coarse-grained $\text{La}_{18}\text{W}_{10}\text{O}_{57}$ ceramic (relative density of 92%), (3) low-density orthorhombic $\beta - \text{La}_2\text{WO}_6$ ceramic (relative density of 69.6%), (4) coarse-grained orthorhombic $\beta - \text{La}_2\text{WO}_6$ ceramic (relative density of 81.4%) and (5) $\text{La}_2\text{W}_{1+x}\text{O}_{6+3x}$ ($x \sim 0.22$) single crystal. **Bottom** (1) low-density hexagonal $\text{La}_{15}\text{Mo}_{8.5}\text{O}_{48}$ nanoceramic (relative density of 64.8%) and (2) low-density tetragonal $\gamma - \text{La}_2\text{MoO}_6$ ceramic (relative density of 69.6%)

Composition, $T_{\text{synthesis}}$	E_a (± 0.01), eV	
	400–600 $^\circ\text{C}$	600–900 $^\circ\text{C}$
$\text{La}_{15}\text{W}_{8.5}\text{O}_{48}$ – 800 $^\circ\text{C}$	Dry air	1.36
	Wet air	1.35
$\text{La}_{18}\text{W}_{10}\text{O}_{57}$ – 1400 $^\circ\text{C}$	Dry air	1.19
	Wet air	1.25
$\beta - \text{La}_2\text{WO}_6$ – 1000 $^\circ\text{C}$	Dry air	1.13
	Wet air	1.03
$\beta - \text{La}_2\text{WO}_6$ – 1400 $^\circ\text{C}$	Dry air	1.16
	Wet air	1.05
$\text{La}_2\text{W}_{1+x}\text{O}_{6+3x}$ ($x \sim 0.22$) single crystal	Dry air	0.89
	Wet air	0.83
$\text{La}_{15}\text{Mo}_{8.5}\text{O}_{48}$ – 600 $^\circ\text{C}$	Dry air	1.35
	Wet air	1.34
$\gamma - \text{La}_2\text{MoO}_6$ – 900 $^\circ\text{C}$	Dry air	1.14
	Wet air	1.09

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